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Oligo(aryleneethynylene)s with Terminal Pyridyl Groups: Synthesis and Length Dependence of the Tunneling-to-Hopping Transition of Single-Molecule Conductances

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Supporting Information

ABSTRACT: The synthesis is reported of a new series of oligo(aryleneethynylene) (OAE) derivatives of up to ca. 6 nm in molecular length (OAE9) using iterative Pd-mediated Sonogashira cross-coupling methodology. The oligo-*p*-phenyl-eneethynylene cores of the molecular wires are functionalized at both termini with pyridyl units for attachment to gold leads. The molecular structures determined by single-crystal X-ray analysis are reported for OAE4, OAE5, OAE7, and OAE8a. The charge transport characteristics of derivatives OAE3–OAE9 in single-molecular junctions have been studied using



the mechanically controlled break junction technique. The data demonstrate that the junction conductance decreases with increasing molecular length. A transition from coherent transport via tunneling to a hopping mechanism is found for OAE wires longer than ca. 3 nm.

KEYWORDS: molecular electronics, synthesis, oligomer, oligo(aryleneethynylene), single-molecule conductance, tunneling, hopping

INTRODUCTION

Molecular electronics^{1–5} is receiving renewed interest due to the intrinsic limitations of silicon-based electronics.⁶ Synthetic chemistry has tailored organic molecules to achieve specific electronic functions, even at the single-molecule level.^{7,8} Terminal anchor groups that have an affinity for gold are attached to the molecules to achieve the in situ assembly of metal–molecule–metal junctions.^{9–11} The electrical conductance of these nanoscale junctions is characterized using a range of techniques, notably, scanning tunneling microscopy (STM)^{12,13} conductive-probe atomic force microscopy (CP-AFM),^{14,15} scanning tunneling microscopy break junctions (STM-BJs),^{16–18} and mechanically controlled break junctions (MCBJs).^{19–21}

Studies that aim to understand how charge transport mechanisms vary with changing the length of the molecular wire^{22,23} have been very limited due to the challenges posed by the synthesis, rigorous purification, and measurement of long molecules anchored between two electrodes. Two distinct charge transport mechanisms have been predicted theoretically and identified experimentally: coherent transport via tunneling or superexchange that dominates in short molecules, and incoherent thermally activated hopping in long molecules.²⁴ Choi et al. reported a series of oligophenyleneimine (OPI) molecules of length ca. 1.5–7 nm with an anchored thiol unit at one terminus and a phenyl unit at the other terminus. Contacting a monolayer of the molecules assembled on gold with a CP-AFM

tip revealed a transition from tunneling to hopping at a molecular length of ca. 4 nm in junctions made of hundreds of molecules.²⁵ Single-molecule measurements were not reported in these studies. Hines et al. reported a transition from tunneling to hopping at a molecular length between 5.2 and 7.3 nm by means of STM-BJ measurements (at the single-molecule level) on four oligofluorene-based molecular wires terminated with thiols.²⁶

Oligo(phenyleneethynylene) (OPE) derivatives are ideal candidates for studies on the correlation of conductance with molecular length. They are highly conjugated, rigid, rodlike molecules whose functional properties can be systematically tuned by chemical synthesis, which makes them a very important class of molecules in materials chemistry and molecular electronics.^{27,28} Short derivatives, typically OPE3 systems, have been widely studied in metal-molecule-metal junctions,²⁹⁻³¹ but there are very few reports of transport studies through longer derivatives. ³² Lu et al. studied a series of OPEs of lengths ca. 1-5nm with amino groups at both termini and probed the charge transport mechanism in monolayers using STM-BJ and CP-AFM methods. A transition from tunneling to hopping was observed at a molecular length of ca. 2.75 nm.³³ Studies on related derivatives incorporating a ferrocene unit in the backbone, using the same contacting techniques, established that ferrocene enhanced the

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Chart 1. Structures of the OAE Derivatives Studied in This Work



OAE3a *n* = 1; R = H **b** *n* = 1; R = OMe **OAE4** *n* = 2; R = OC₆H₁₃ **OAE5** *n* = 3; R = OC₆H₁₃



OAE6 *n* = 2 **OAE7** *n* = 3



OAE8a *n* = 2; R = H **b** *n* = 2; R = OC₆H₁₃ **OAE9** *n* = 3; R = OC₆H₁₃

Scheme 1. Synthesis of OAE4 and $OAE9^a$



"Reagents and conditions: (i) 2-methyl-3-butyn-2-ol, $[PdCl_2(PPh_3)_2]$, CuI, $(i-Pr)_2NH$, 20 h, 20 °C, 95% yield; (ii) NaOH, toluene, 6 h, 70 °C, 65% yield; (iii) 4, $[Pd(PPh_3)_4]$, CuI, THF, Et₃N, 18 h, 50 °C, 95% yield; (iv) NaOH, toluene, 22 h, 100 °C, 36% yield; (v) 7, $[Pd(PPh_3)_4]$, CuI, THF, Et₃N, 23 h, 20 °C, 88% yield; (vi) 8, $[Pd(PPh_3)_4]$, CuI, THF/Et₃N, 18 h, 20 °C, 60% yield; (vii) 1, $[Pd(PPh_3)_4]$, CuI, THF, Et₃N, 15 h, 20 °C, 21% yield.

conductance in both the tunneling and the hopping regimes.³⁴ Single-molecule measurements were not reported in these studies.^{33,34}

Terminal 4-pyridyl (PY) units anchor molecules effectively to gold substrates, and unlike thiols,^{35,36} they are stable under ambient conditions and no protecting group is needed.^{37–40} We

Scheme 2. Synthesis of OAE5^a



"Reagents and conditions: (i) NaOH, toluene, 12 h, 100 °C, 82% yield; (ii) **3**, [Pd(PPh₃)₄], CuI, THF, Et₃N, 18 h, 20 °C, 47% yield; (iii) NaOH, toluene, 23 h, 100 °C, 70% yield; (iv) 7, [Pd(PPh₃)₄], CuI, THF, Et₃N, 20 h, 20 °C, 68% yield.

	Table 1	1. Characteristic	Parameters of t	he UV/	′vis Abso	rption and	Transpor	t Measurements
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molecule	$\lambda_{ m max}/ m nm^a$	L/nm^b	$G^*/G_0 1D^c$	$G/G_0 2D^d$	Δz^{*^e}	z^{*f}	JFP/% ^g
OAE3a	319	1.6	10 ^{-4.5}	$10^{-5.0} - 10^{-4.0}$	1.08	1.58	100
OAE3b	306, 372	1.6	10 ^{-4.5}	$10^{-5.0} - 10^{-4.0}$	1.00	1.50	100
OAE4	320, 401	2.3	10 ^{-5.5}	$10^{-6.0} - 10^{-5.0}$	1.88	2.38	100
OAE5	324, 417	3.0	10 ^{-6.7}	$10^{-7.0} - 10^{-6.2}$	1.92	2.42	71
OAE6	331, 409	3.7	$10^{-6.8}$	$10^{-7.3} - 10^{-6}$	2.22	2.72	18
OAE7	339, 428	4.4	$10^{-6.9}$	$10^{-7.0} - 10^{-6.5}$	2.31	2.81	18
OAE8b	335, 435	5.1	10 ^{-6.8}	$10^{-7.2} - 10^{-6}$	1.78	2.28	20
OAE9	335, 439	5.8	10 ^{-6.9}	$10^{-7.5} - 10^{-6}$	1.36	1.86	17

^{*a*}Maxima of the characteristic UV/vis absorption in CH₂Cl₂ solution; ^{*b*}Molecular length *L*, which is defined as the distance between the center of the nitrogen anchor atom at one end of a fully extended isolated molecule to the center of the anchor atom at the other end. The lengths of the molecules were calculated by ACD/ChemSketch and are in very close agreement with the N…N distances obtained by single-crystal X-ray diffraction for **OAE4**, **OAE5**, **OAE7** and **OAE8a** (see the Supporting Information). ^{*c*}Most probable molecular junction conductance as estimates from Gaussian fits to the experimentally obtained 1D conductance histograms. ^{*d*}Conductance range of molecular junctions extracted from the 2D conductance vs relative displacement histograms. ^{*c*} Δz^* as most probable relative junction elongation (displacement). ^{*f*} $z^* = \Delta z^* + 0.5$ as absolute junction elongation (displacement).

have shown that, for tolane derivatives, pyridyl exhibits excellent anchoring performance with high conductance and 100% of molecular junctions formed.⁴¹ Therefore, pyridyl units are used as anchors in the present series of molecules, in preference to amine or thiol analogues.

The motivation for this work is to study the length dependence of the *single-molecule* conductance of oligoaryleneethynylene (OAE) derivatives comprising OPE cores terminated at both ends with pyridyl groups. This study is clearly distinct from previous work on OPI²⁵ and OAE^{33,34} derivatives that concerned monolayers comprising hundreds of molecules in closely packed arrangements. The benefit of single-molecule studies is that artifacts arising from intermolecular interactions (e.g., $\pi - \pi$ stacking) that are well-known in OAE derivatives²⁷ are absent. We describe the synthesis and characterization of derivatives **OAE3–OAE9** of molecular length (i.e., intramolecular N···N distance) ca. 2–6 nm. The charge transport characteristics of these OAE derivatives in single-molecular junctions have been studied using the MCBJ technique in solution and under environmentally controlled conditions.

RESULTS AND DISCUSSION

Synthesis. The OAE derivatives studied in this work are shown in Chart 1. The synthetic strategy uses iterative Pd-catalyzed Sonogashira cross-coupling reactions⁴² of new, linearly conjugated phenyleneethynylene building blocks. The retrosynthetic disconnections of the molecules are based on (i) commercial availability of starting materials, (ii) high yielding reactions, (iii) and the synthesis of intermediate reagents that

could be easily purified, thereby facilitating the final step to the target OAE derivatives. For the derivatives OAE4-OAE9, hexyloxy substituents are attached to the inner phenyl rings to give the molecules good solubility in organic solvents. Two OAE3 derivatives were studied (OAE3a and OAE3b) to confirm that alkoxy substituents do not affect the conductance behavior, as was reported previously for thiol-terminated OPE3 analogues.³⁰ Two OAE8 derivatives were synthesized: OAE8a has only very limited solubility, and although an X-ray crystal structure was obtained, it could not be purified to the required level for conductance measurements. The analog OAE8b with additional hexyloxy side chains in the backbone has improved solubility and was fully characterized. A key aspect of our strategy is the use of 2-hydroxypropyl protecting groups for the intermediate terminal alkynes, e.g., compounds 2 and 5 (Scheme 1). A benefit of this protecting group (compared to trialkylsilyl, which is more commonly used)²⁸ is that its high polarity facilitates the separation of any unreacted starting material or byproducts that do not contain this unit. Deprotection with loss of acetone is easily accomplished using sodium hydroxide in toluene.⁴³ A notable feature of the deprotection reactions is that, for bis-protected derivatives 2 and 5, the major product can be exclusively monodeprotected $(2 \rightarrow 3)$ or bis-deprotected $(5 \rightarrow$ 6) depending on the temperature and the length of time of the reaction. As representatives of the series, the syntheses of OAE4 and OAE9 are shown in Scheme 1, and OAE5 in Scheme 2. The syntheses of the other OAEs are described in the Supporting Information. All the OAE derivatives were characterized by ¹H NMR, ¹³C NMR, and mass spectra that confirmed their expected

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structures and established their high level of purity. Single-crystal X-ray structures obtained for OAE4, OAE5, OAE7, and OAE8a give the molecular lengths and solid-state conformations of the molecules. UV–visible absorption spectra in solution (Table 1) show that, for OAE3-OAE9, there is extended conjugation throughout the backbone. The saturation length has not been reached in the present series. These crystallographic and spectroscopic characterization data are discussed in detail in the Supporting Information.

Conductance Measurements. The charge transport characteristics of the derivatives **OAE3–OAE9** in single-molecular junctions were investigated using the mechanically controlled break junction (MCBJ) technique. Key parameters are listed in Table 1. Molecular junctions were formed by opening and closing a nanogap between two gold electrodes in a solution of 1,3,5-trimethylbenzene (TMB) and tetrahydrofuran (THF), 4:1 (v/v), containing typically 0.1 mM of the target molecules. Details of the experimental setup were described in our previous publications, 41,44 and are briefly summarized in the Experimental Section.

Figure 1 displays typical conductance (G) versus distance (Δz) stretching traces, as plotted in a semilogarithmic scale, and



Figure 1. Typical conductance vs distances traces recorded in TMB/ THF in the absence (black lines) and in the presence of 0.1 mM **OAE4** (blue lines) or **OAE7** (red lines) in an MCBJ setup at 0.10 V bias voltage, stretching rate of 2 nm s⁻¹.

recorded for 0.1 mM OAE4 or OAE7 in TMB/THF using the MCBJ technique. For reference, we also plotted two traces (black curves) representing the OAE-free solution, which reveal classical tunneling characteristics, i.e., an exponential decrease of the conductance upon junction elongation. The traces recorded for OAE4 and OAE7, which are characteristic representatives of the studied pyridyl-terminated OAE family, are more complex. After the formation of the contact between the two gold leads, the nanoscale constriction was stretched with a typical rate of 2 nm s⁻¹. All curves show initially a steplike decrease of the conductance from 10 G_0 up to 1 G_0 , with G_0 = $2e^2/h$ being the fundamental quantum conductance. Opening the gap results in an elongation of the gold-gold junction and decreases the number of gold atoms in the constriction, which causes the conductance to change up to 1 G_0 , where the contact between the two gold leads in the MCBJ consists of only one gold atom. Subsequently, an abrupt decrease of the conductance over several orders of magnitude up to 10^{-5} G₀ occurs, which is assigned to the "jump out of atomic contact".⁴⁵ The gold–gold contact breaks, and OAE molecules from the solution or already adsorbed at one of the two metal leads bridge the gap. This

process is reflected in additional features in the log(G/G_0) vs Δz stretching traces. The examples plotted in Figure 1 show welldeveloped plateaus at $G < 10^{-5} G_0$. These plateaus represent stable molecular junctions "electrode-OAE-electrode". We notice that these traces show a distinct monotonic decrease in the conductance with stretching distance. Upon further pulling, a second decrease in conductance occurs, which is assigned to the rupture of the contact between the molecular wire and the gold leads. ^{31,46} The conductance finally reaches the noise level of $G \leq$ 10^{-8} G₀ upon the complete breaking of the molecular junction. We note that the plateau length increases with molecular length L, as illustrated in Figure 1 for OAE4 and OAE7 as typical examples. We also observed that rupture of the molecular junction is less steep for junctions formed with the longer OAE molecules. The overall shape of the stretching traces as recorded for each molecule is qualitatively similar, but varies in detail. Therefore, a careful statistical analysis of a large number of individual traces is required to extract representative results.

Several thousands of individual conductance versus relative displacement traces (G vs Δz) were recorded and analyzed further by constructing all-data-points histograms without any data selection to extract statistically significant results from the different junction configurations. Figure 2 displays the



Figure 2. All-data-points 1D conductance histograms of OAE derivatives as constructed from 2000 individual conductance vs relative distance traces as plotted in Figure 3.

corresponding one-dimensional (1D) histograms of eight OAEs in a semilogarithmic scale as constructed from 2000 experimental conductance versus distance traces for each compound. The sharp peaks around 1 G_0 represent the conductance of a single atom gold-gold contact. The prominent peaks between $10^{-7} G_0 < G/G_0 < 10^{-4} G_0$ represent molecular features. The 1D histograms reveal one pronounced molecular feature for each OAE derivative. The conductance peaks at G < $10^{-7.5}$ G₀, which were also observed in OAE-free solutions, are caused by instrumental noise.^{41,44} The statistically most probable conductances of molecular junctions formed by the respective OAEs as trapped between two gold contacts were obtained by fitting Gaussians to the characteristic maxima in the conductance histograms. The results are compiled in Table 1. The data presented in Figures 2 and 3 are summarized in Figure 4. We note that rather similar conductance values were found for molecular concentrations ranging between 10^{-6} M up to 10^{-4} M, which supports their assignment to single-molecular junction conductances. The single junction conductances decrease with increasing molecular length L from $10^{-4.5} G_0$ for OAE3 to $10^{-6.7}$ G_0 for OAE5, by 3 orders of magnitude. However, this trend levels off for longer molecular wires. The junction conductances of OAE5 to OAE9 reveal rather small changes spanning from $10^{-6.7}$ to $10^{-6.9}$ G₀, respectively. We also note that the width of the molecular conductance features increases with molecular

length, which most probably reflects the increasing number of molecular conformations in the nanoscale junctions.

The above analysis was extended by constructing twodimensional (2D) conductance vs relative displacement Δz histograms. Figure 3A,B shows typical results for **OAE4** and



Figure 3. (A, B) 2D conductance–relative displacement histograms from MCBJ-BJ experiments for OAE4 (A) and OAE7 (B). (C, D) Relative displacement (Δz) distribution from MCBJ-BJ experiments for OAE4 (C) and OAE7 (D). The relative displacement histograms are obtained from conductance traces between 0.1 G_0 and 10⁻⁷ G_0 . The envelope traces represent Gaussian fits to the molecular junction contribution and to direct tunneling, labeled as M and T, respectively.

OAE7. The corresponding graphs of the other molecules are summarized in the Supporting Information. The 2D conductance histograms were obtained as follows.^{41,47,48} First, we normalized all individual conductance traces to a common distance scale by assigning the relative displacement $\Delta z = 0$ at G = 0.7 G_0 . This procedure is justified by the sharp drop in conductance around 1 G_0 . The conductance versus relative displacement histogram was then constructed by counting the occurrences of $[\log(G/G_0), \Delta z]$ pairs in a 2D field. The 2D histograms of OAE4 and OAE7 show features of gold quantum contacts around $G \ge 1$ G_0 and a second cloudlike pattern in [0.5 nm < Δz < 1.8 nm, $10^{-6.2}$ G_0 < G < $10^{-4.8}$ G_0] centered at G = $10^{-5.5} G_0$ (**OAE4**) and in [0.5 nm < Δz < 2.2 nm, $10^{-7.0} G_0$ < G < 10^{-6.5} G_0] centered at $G = 10^{-6.9} G_0$ (**OAE7**). We attribute the latter to the formation of single-molecule junctions. The corresponding characteristic conductance data are rather close to the positions of conductance peaks in the 1D histograms. Similar conclusions were also obtained for the other OAE derivatives (see the Supporting Information). We also note that all 2D conductance histograms reveal a small, but distinct, decrease of the single junction conductance G with increasing displacement Δz . The corresponding intervals are given in Table 1 for all the compounds investigated. Finally, we comment that the data cloud below $10^{-7.5} G_0$ represents the noise level.

Analyzing the evolution of a molecular junction upon stretching provides additional information about the stability and junction formation probability. We constructed relative displacement (Δz) histograms⁴¹ by calculating the displacement from the relative zero position at 0.7 G_0 to the end of the molecular conductance region, just before the onset of the noise level. Figure 3C,D displays characteristic displacement histograms of OAE4 and OAE7. The plot of OAE4 reveals a uniform normal distribution with a well-defined maximum. No stretching traces shorter than 1.2 nm were observed, indicating that no significant contribution due to direct tunneling (T) through the solution exists. Therefore, we conclude that the molecular junction formation probability of OAE4 approaches 100%. The single maximum in Figure 3C represents the most probable relative characteristic stretching distance $\Delta z^* = 1.88$ nm, which may be considered as a measure of the most probable characteristic plateau length of an OAE4-type molecular junction.⁴¹ The most probable absolute displacement z^* in an experimental molecular junction formed between two gold tips and **OAE4** is obtained by adding the snap-back distance Δz_{corr} to the relative displacement: $z^* = \Delta z^* + \Delta z_{corr}$. Referring to our previous work with pyridyl-terminated rigid rodlike tolanes, we used $\Delta z_{corr} = (0.5 \pm 0.1)$ nm,⁴¹ which leads to $z^* = 2.38$ nm for **OAE4**. (This value is close to the molecular length *L* of **OAE4**.) Similar trends were found with OAE3 (c.f. data in Table 1).

The 1D displacement histograms of the longer OAEs are more complex. Figure 3D illustrates data of OAE7 as an example. In addition to a well-developed molecular feature with a maximum at $\Delta z^* = 2.31$ nm, which gives $z^* = 2.81$ nm as the most probable absolute displacement, we observed a second peak with a maximum around 1.2 nm. This peak originates from an increasing number of individual conductances versus distance traces without the formation of a molecular junction. In other words, it represents contributions from direct tunneling. The area ratio between the molecular contribution and the total data density in the histogram plotted in Figure 3D leads to a junction formation probability of 18% for OAE7. A similar analysis of our experiments with OAE6 to OAE9 reveals the following trends: (i) The molecular junctions break prior to a fully extended configuration; (ii) the molecular features in the 1D displacement histograms broaden; and (iii) the junction formation probability decreases considerably with increasing molecular length L. These observations demonstrate that the larger number of molecular conformations and/or the increasing number of hexyloxy sidechains of the longer OAEs appear to hamper the formation of stable (single) molecular junctions. Changes in the molecular structure near the pyridyl binding site may play a role. For OAE3-OAE5, the phenyl rings adjacent to the pyridyls are disubstituted with alkoxy groups, whereas for OAE6-OAE9 these rings are unsubstituted. The latter might strengthen the π interaction of the molecule with the gold substrate, and therefore, an additional energy might be needed to lift the respective OAEs from the substrate surface. This extra energy threshold might not always be reached so that the percentage of stretching events that are unsuccessful (no molecular junctions being formed) is larger, which leads to a lower junction formation probability. This process is currently being studied by molecular dynamics simulations in combination with electromechanical forcedistance and conductance-distance measurements. However, this topic is beyond the scope of this article.

Figure 4 displays the dependence of the most probable single junction conductances of the pyridyl-terminated OAE family on molecular length *L*. We found two distinctly different regions. For the shorter molecules, from **OAE3** to **OAE5**, the conductance decreases exponentially according to eq $1^{23,31}$

$$G = G_{\rm c} {\rm e}^{-\rho L} \tag{1}$$

with an attenuation constant $\beta = (3.3 \pm 0.1) \text{ nm}^{-1}$ and a contact conductance per pyridyl terminating group, $G_c = (0.54 \pm 0.07) \mu \text{S}$ at L = 0, which leads to a contact resistance $R_c = 1/G_c = (1850 \pm 232) \text{ k}\Omega$. Similar values of β were reported for other rigid-



Figure 4. Semilogarithmic plot of the most probable single-molecule junction conductances (individual data points as obtained from the analysis of the 1D conductance histograms) and of the conductance range (bare symbols as extracted from the analysis of the 2D conductance histograms) in units of $log(G/G_0)$ versus the molecular length for a family of pyridyl-terminated OAE derivatives. The inset represents the junction formation probability (JFP) dependent on the molecular length L. All numerical data are compiled in Table 1

rodlike single molecular wires or assembled monolayers, such as dithiol-terminated OPEs $(3.4 \pm 0.1 \text{ nm}^{-1})$,³¹ oligophenyleneimines (OPI, $\beta = 3.0 \text{ m}^{-1}$),²⁵ oligophenylenedithiols (PP, $\beta = (3.5-5.0) \text{ nm}^{-1}$),^{49,50} or pyridyl-terminated oligoynes (OY, $\beta =$ (3.1 ± 0.4) nm⁻¹).⁵¹ Smaller values of β were reported for OPEs with amine anchoring groups (OPE, $\beta = 2.0 \text{ nm}^{-1}$),³³ oligoflourenes (OF, $\beta = (2.06 \pm 0.09 \text{ m}^{-1}))$,²⁶ oligopheny-lene-vinylenes (OPV, $\beta = (1.7-1.8) \text{ nm}^{-1}$),^{46,52} oligothiophenes (OT, $\beta = \sim 1 \text{ nm}^{-1}$),⁵³ and for metal-porphyrin containing molecular wires (P, $\beta \ll 0.1 \text{ nm}^{-1}$).⁵⁴

The contact resistance $R_c = 1850 \text{ k}\Omega$ per pyridyl anchoring group for the family of OAE wires studied in this work is larger than $R_c = 205 \text{ k}\Omega$, which is the corresponding value for pyridylterminated oligoynes.⁵¹ A similar trend is observed with the dithiol-terminated OPEs $(40 \text{ k}\Omega)^{31}$ and oligoynes (~3.2 k Ω).⁵¹ These data demonstrate that the values are not only determined by the molecular anchoring site but also by the coupling of the anchor group to the wire backbone.

Comparing the results of the single-molecule conductance measurements for the short OAEs OAE3-OAE5 with the literature data above, and, in particular, with results reported for pyridyl-terminated oligoynes⁵¹ and thiol-terminated OPEs,³¹ as well as the excellent fit of eq 1 to the experimental data, demonstrates that electron transport across the OAE3-OAE5 wires is controlled by coherent transport via tunneling or superexchange.

A distinctly different behavior is found for the longer derivatives OAE6-OAE9. The most probable single junction conductances are still exponentially dependent on molecular length L. However, the fit of eq 1 to the experimental data gives an attenuation constant $\beta = (0.16 \pm 0.08) \text{ nm}^{-1}$, which is significantly smaller as compared to the $\beta = (3.3 \pm 0.01)$ nm⁻¹, the values extracted from the analysis of the shorter OAEs. Similar small β values were reported for longer amine-terminated OPEs ($\beta \sim 0.3 \text{ nm}^{-1}$),³³ thiol-terminated oligofluorenes ($\beta \ll$ 0.1 nm⁻¹),²⁶ and longer oligophenylenimines ($\beta \sim 0.9 \text{ nm}^{-1}$).²³ Comparing these results suggests that transport through the longer pyridyl-terminated OAEs is characterized by a thermally activated hopping mechanism.²⁴ Transition from tunneling to hopping was found at a molecular length of ~3 nm for the present system.

Single-molecule studies of length dependence have not been reported previously for OAE systems. CP-AFM experiments,

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which involve contacting hundreds of molecules within selfassembled monolayers, revealed the transition from tunneling to hopping at a molecular length of ca. 2.75 nm for a series of amineterminated OPEs³³ and ca. 4 nm for thiol-anchored oligophenyleneimines.²⁵ Single-molecule conductance experiments employing the STM-BJ techniques reported characteristic transition lengths of 5.6-6.8 nm for polythiophenes⁵⁵ and 5.2-7.3 nm for oligofluorene-based molecular wires,²⁶ both terminated with thiol anchors. We note that, for the oligofluorene series, the experiments are very limited within the length range investigated.²⁶ This comparison demonstrates convincingly that the transition length from tunneling to transport controlled by hopping depends critically on the structure of the molecular backbone. However, the existing limited database of single-molecular junction studies does not yet allow general trends to be extracted. Comparative studies with well-defined molecular wires, not only of known and tunable length but, in particular, of molecular conformation, are essential to achieve this goal.

CONCLUSIONS

We have presented the synthesis of a family of new linear OAE derivatives with terminal pyridyl anchor groups of molecular length ca. 2-6 nm. The charge transport properties of these molecular wires were characterized by MCBJ experiments. Single-molecule junction conductance measurements reveal a transition from coherent transport via tunneling, which dominates in shorter molecules, to incoherent hopping in longer molecules. The transition occurs at a molecular length $L \sim 3.0$ nm and for conductances below $10^{-6.5}$ G₀. Increasing the molecular length L leads also to a reduction of the probability for molecular junction formation from 100% to values below 20%. This trend is attributed to the larger number of molecular conformations and the increasing number of hexyloxy sidechains, which appear to hamper the formation of stable OAEtype (single) molecular junctions. A comparison of our data with literature data for other oligomer systems establishes that the details of the tunneling-to-hopping transition depend on the nature of the molecular wire backbone as well as its coupling to the respective anchoring sites. This work represents the first studies on length dependence of conductance of OAE derivatives at the single-molecule level. However, unambiguous and more general correlations between structure and transport characteristics cannot yet be established due to the limited number of reliable single junction conductance studies with long molecular wires of well-defined structure, in particular, molecular conformation. This topic comprises a major challenge for future research in synthetic and materials chemistry.

EXPERIMENTAL SECTION

Characterization Data for the OAE Derivatives. OAE3a. This compound was synthesized following the literature route, and the characterization data are in agreement with those reported. 40

OAE3b. mp 189.1–190.2 °C. ¹H NMR (400 MHz, CDCl₃) δ 8.63 (dd, J = 4.5, 1.6 Hz, 4H), 7.44 (dd, J = 4.5, 1.6 Hz, 4H), 7.07 (s, 2H), 3.94 (s, 6H); ¹³C NMR (CDCl₃, 101 MHz) 154.57, 150.04, 131.49, 125.72, 116.19, 113.63, 92.71, 90.21, 56.78; HR-MS (ASAP+) m/z Calcd for C₂₂H₁₆N₂O₂ ([M]⁺ 340.1212, found *m*/*z*: [M]⁺ 340.1229.

OAE4. mp 107.0–107.5 °C. ¹H NMR (700 MHz, CDCl₃) δ 8.60 (d, J = 5.6 Hz, 4H), 7.37 (d, J = 5.6 Hz, 4H), 7.02 (s, 2H), 7.01 (s, 2H), 4.03 (m, 8H), 1.84 (m, 8H), 1.52 (m, 8H), 1.34 (m, 16H), 0.89 (m, 12H); $^{13}\mathrm{C}\,\mathrm{NMR}\,(151\,\mathrm{MHz},\mathrm{CDCl}_3)\,\delta$ 153.98, 153.45, 149.61, 131.69, 125.41, 117.14, 116.96, 115.33, 112.66, 91.94, 91.75, 90.80, 69.74, 69.51, 31.58, 31.54, 29.25, 29.23, 25.71, 25.64, 22.62, 14.01, 13.99; HR-MS (ASAP+)

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m/z Calcd for C₅₂H₆₅N₂O₄ [M + H]⁺ 781.4944, found m/z: [M + H]⁺ 781.4937.

OAE5. mp 127.2–128.2 °C. ¹H NMR (700 MHz, CDCl₃) δ 8.59 (d, *J* = 5.7 Hz, 4H), 7.36 (d, *J* = 5.7 Hz, 4H), (7.01 (m, 6H), 4.02 (m, 12H), 1.84 (m, 12H), 1.51 (m, 12H), 1.33 (m, 24H), 0.88 (m, 18H); ¹³C NMR (176 MHz, CDCl₃) δ 153.99, 153.53, 153.41, 149.63, 131.67, 125.40, 117.24, 117.18, 116.95, 115.54, 114.27, 112.49, 92.05, 91.87, 91.31, 90.81, 69.76, 69.66, 69.50, 31.59, 31.58, 31.54, 29.27, 29.24, 29.23, 25.70, 25.66, 25.64, 22.61, 14.00, 13.98; HR-MS (ASAP+) *m/z* Calcd for C₇₂H₉₃N₂O₆ [M + H]⁺ 1081.7034, found *m/z*: [M + H]⁺ 1081.7070.

OAE6. mp 214.2–215.2 °C. ¹H NMR (600 MHz, CDCl₃) δ 8.62 (bs, 4H), 7.52 (bs, 8H), 7.39 (d, *J* = 4.2 Hz, 4H), 7.01 (s, 2H), 7.00 (s, 2H), 4.03 (m, 8H), 1.85 (m, 8H), 1.53 (m, 8H), 1.34 (m, 16H), 0.89 (m, 12H); ¹³C NMR (151 MHz, CDCl₃) δ 153.72, 153.49, 149.53, 131.78, 131.54, 131.43, 125.58, 124.44, 121.64, 117.02 (2 overlapping peaks), 114.68, 113.51, 94.30, 93.85, 91.64, 88.61, 88.30, 69.72, 69.55, 31.60, 31.59, 29.30, 29.26, 25.74, 25.65, 22.62, 14.02; HR-MS (ASAP+) *m*/*z* Calcd for C₆₈H₇₃N₂O₄ [M + H]⁺ 981.5570, found *m*/*z*: [M + H]⁺ 981.5566.

OAE7. mp 203.5–204.2 °C. ¹H NMR (600 MHz, CDCl₃) δ 8.61 (d, *J* = 6.0 Hz, 4H), 7.52 (s, 8H), 7.39 (d, *J* = 6.0 Hz, 4H), 7.01 (m, 3H), 4.03 (m, 12H), 1.84 (m, 12H), 1.52 (m, 12H), 1.34 (m, 24H), 0.89 (m, 18H); ¹³C NMR (151 MHz, CDCl₃) δ 153.73, 153.50, 153.47, 149.49, 131.79, 131.54, 131.50, 125.55, 124.47, 121.61, 117.23, 117.04, 117.01, 114.78, 114.28, 113.42, 94.26, 93.92, 91.78, 91.45, 88.64, 88.27, 69.73, 69.67, 69.54, 31.61, 31.59, 29.30, 29.29, 29.26, 25.75, 25.67, 25.65, 22.62, 14.02; HR-MS (ASAP+) *m*/*z* Calcd for C₈₈H₁₀₁N₂O₆ [M + H]⁺ 1281.7700, found *m*/*z*: [M]⁺ 1281.7615.

OAE8b. mp 199.5–200.6 °C. ¹H NMR (600 MHz, CDCl₃) δ 8.61 (d, J = 5.2 Hz, 4H), 7.53 (s, 8H), 7.38 (d, J = 5.2 Hz, 4H), 7.02 (m, 8H), 4.04 (m, 16H), 1.86 (m, 16H), 1.53 (m, 16H), 1.35 (m, 32H), 0.89 (m, 24H); ¹³C NMR (151 MHz, CDCl₃) δ 153.74, 153.52, 153.50, 153.48, 149.72, 131.78, 131.54, 131.26, 125.48, 124.44, 121.67, 117.26, 117.24, 117.06, 117.02, 114.81, 114.40, 114.21, 113.42, 94.28, 93.68, 91.83, 91.61, 91.44, 88.65, 88.34, 69.74, 69.68, 69.66, 69.54, 31.62, 31.61, 31.60, 29.32, 29.30, 29.27, 25.76, 25.68, 25.57, 22.64, 14.03; HR-MS (ASAP+) m/z calcd for C₁₀₈H₁₂₈N₂O₈ [M]⁺ 1580.9671, found m/z: [M]⁺ 1580.9725.

OAE9. mp 202.1–203.5 °C. ¹H NMR (400 MHz, CDCl₃) δ 8.62 (d, *J* = 6.0 Hz, 4H), 7.54 (bs, 8H), 7.40 (d, *J* = 6.0 Hz, 4H), 7.02 (m, 10H), 4.03 (m, 20H), 1.86 (m, 20H), 1.53 (m, 20H), 1.34 (m, 40H), 0.89 (m, 30H); ¹³C NMR (101 MHz, CDCl₃) δ 153.95, 153.74, 153.71 (two overlapping peaks), 153.69, 149.91, 132.04, 131.80, 131.59, 125.77, 124.68, 121.89, 117.98, 117.44, 117.22, 115.01, 114.79, 114.61, 114.50, 114.38, 113.61, 112.82, 94.52, 94.01, 92.07, 91.87, 91.80, 91.66, 88.88, 88.56, 69.95, 69.88, 69.75, 31.87, 31.85, 31.78, 29.53, 29.50, 26.00, 25.92, 25.86, 22.89, 14.29; MS (ASAP⁺) *m*/*z*: 1883.1 ([M + H]⁺, 100%).

Mechanically Controlled Break Junction (MCBJ) Measurements. The transport characteristics in single-molecule junctions were studied by MCBJ measurements in solution at room temperature. The latter contained typically 0.1 mM of the OAE-type molecules in a mixture of 1,3,5-trimethylbenzene (TMB; Aldrich, p.a.) and tetrahydrofuran (THF; Aldrich, p.a), 4:1 (v/v).

The MCBJ experiments are based on the formation and breaking of a nanogap between a notched, freely suspended gold wire (0.1 mm diameter, 99.999%, Goodfellow), fixed on spring steel sheets (10 mm × 30 mm, thickness 0.25 mm) with a two-component epoxy glue (Stycast 2850 FT with catalyst 9). The sample sheets were fixed between two holders. A Kel-F liquid cell with a Kalrez O-ring was mounted onto the sheet. During the measurements, the steel sheet could be bent with a pushing rod, which was controlled by a combination of a stepper motor and a piezo stack. The bending was initialized by the stepper motor. Once the measured current decreased to a value corresponding to 15 $G_{0\nu}$ the stepper motor stops to move, and the piezo stack was activated. This strategy reduced significantly noise contributions from the operation of the stepper motor. The movement of the piezo stack controlled the breaking and the reformation of nanoscale contacts, typically in the range between the noise threshold $(G < 10^{-8} G_0)$ and a high conductance limit, which was set to 10 G₀. Molecular junctions could form upon breaking the gold-gold nanocontacts. The entire cycle was

repeated more than 2000 times to obtain statistically relevant data. In the MCBJ setup, the current could be recorded as a feedback signal at a given bias voltage (typically between 0.020 and 0.200 V). The two ends of the "broken wire" were taken as working electrodes WE1 and WE2.

The MCBJ unit is controlled by a lab-built bipotentiostat with two bipolar tunable logarithmic I - V converters as current measuring units, which are operated by a custom-designed microcontroller. The system provides three analog signals: the potential of WE1, the voltage difference between the two working electrodes WE1 and WE2 (bias voltage V_{bias}), driving the current through the two gold electrodes for the conductance measurements, and the voltage output of the piezo stack in the range of 0-50 V, allowing the displacement of the piezo stack up to 8 μ m with rates ranging from 3 to 3000 nm s⁻¹. The latter translates into lateral pulling (pushing) rates between the two gold leads of 0.1–100 nm s⁻¹. The distance between the two gold electrodes in the MCBJ setup was calibrated with complementary STM-BJ experiments assuming that the tunneling decay is identical under the same experimental conditions. Further technical details of the MCBJ setup are reported by Hong et al. in refs 41 and 44.

ASSOCIATED CONTENT

Supporting Information

Details of molecular synthesis and characterization, X-ray crystallographic data (CCDC 951746–951749), UV–vis absorption spectra, and single-junction conductance measurements (PDF). This material is available free of charge via the Internet at http://pubs.acs.org.

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Notes

The authors declare no competing financial interest.

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REFERENCES

(1) Wassel, R. A.; Gorman, C. B. Angew. Chem., Int. Ed. 2004, 33, 5120–5123.

(2) Joachim, C.; Gimzewski, J. K.; Aviram, A. Nature 2000, 408, 541–548.

(3) Cuevas, J. C.; Scheer, E. Molecular Electronics: An Introduction to Theory and Experiment; World Scientific: Singapore, 2010.

- (4) Mann, B.; Kuhn, H. J. Appl. Phys. 1971, 42, 4398-4405.
- (5) Aviram, A.; Ratner, M. A. Chem. Phys. Lett. 1974, 29, 277-283.
- (6) Mack, C. A. IEEE Trans. Semicond. Manuf. 2011, 24, 202-207.
- (7) Weibel, N.; Grunder, S.; Mayor, M. Org. Biomol. Chem. 2007, 5, 2343–2353.
- (8) James, D. K.; Tour, J. M. Aldrichimica Acta 2006, 39, 47-56.
- (9) James, D. K.; Tour, J. M. Chem. Mater. 2004, 16, 4423-4435.
- (10) McCreery, R. L. Chem. Mater. 2004, 16, 4477-4496.

(11) Chen, F.; Hihath, J.; Huang, Z. F.; Li, X. L.; Tao, N. J. Annu. Rev. Phys. Chem. 2007, 58, 535-564.

(12) Donhauser, Z. J.; Mantooth, B. A.; Kelly, K. F.; Bumm, L. A.; Monnell, J. D.; Stapleton, J. J.; Price, D. W.; Rawlett, A. M.; Allara, D. L.; Tour, J. M.; Weiss, P. S. *Science* **2001**, *292*, 2303–2307.

(13) Repp, J.; Meyer, G.; Paavilainen, S.; Olsson, F. E.; Persson, M. Science **2006**, 312, 1196–1199.

Chemistry of Materials

(14) Cui, X. D.; Primak, A.; Zarate, X.; Tomfohr, J.; Sankey, O. F.; Moore, A. L.; Moore, T. A.; Gust, D.; Harris, G.; Lindsay, S. M. *Science* **2001**, *294*, 571–574.

(15) Fan, F. R. F.; Yang, J. P.; Cai, L. T.; Price, D. W.; Dirk, S. M.; Kosynkin, D. V.; Yao, Y. X.; Rawlett, A. M.; Tour, J. M.; Bard, A. J. *J. Am. Chem. Soc.* **2002**, *124*, 5550–5560.

(16) Xu, B. Q.; Tao, N. J. J. Science 2003, 301, 1221-1223.

(17) Venkataraman, L.; Klare, J. E.; Nuckolls, C.; Hybertsen, M. S.; Steigerwald, M. L. *Nature* **2006**, *442*, 904–907.

(18) Haiss, W.; van Zalinge, H.; Higgins, S. J.; Bethell, D.; Hobenreich, H.; Schiffrin, D. J.; Nichols, R. J. *J. Am. Chem. Soc.* **2003**, *125*, 15294–15295.

(19) Reed, M. A.; Zhou, C.; Muller, C. J.; Burgin, T. P.; Tour, J. M. Science 1997, 278, 252–254.

(20) Smit, R. H. M.; Noat, Y.; Untiedt, C.; Lang, N. D.; van Hemert, M. C.; van Ruitenbeek, J. M. *Nature* **2002**, *419*, 906–909.

(21) Wu, S. M.; Gonzalez, M. T.; Huber, R.; Grunder, S.; Mayor, M.; Schonenberger, C.; Calame, M. Nat. Nanotechnol. **2008**, *3*, 569–574.

(22) Lui, H.; Yu, C.; Gao, N.; Zhao, J. ChemPhysChem **2010**, *11*, 1895–1902.

(23) Luo, L.; Choi, S. H.; Frisbie, C. D. Chem. Mater. 2011, 23, 631–645.

(24) Davis, W. B.; Svec, W. A.; Ratner, M. A.; Wasielewski, M. R. *Nature* **1998**, *396*, 60–63.

(25) Choi, S. H.; Kim, B.; Frisbie, C. D. Science 2008, 320, 1482–1486.

(26) Hines, T.; Diez-Perez, I.; Hihath, J.; Liu, H.; Wang, Z.-S.; Zhao, J.; Zhou, G.; Müllen, K.; Tao, N. J. Am. Chem. Soc. **2010**, 132, 11658–11664.

(27) Bunz, U. H. F. Chem. Rev. 2000, 100, 1605-1644.

(28) Jenny, N. M.; Mayor, M.; Eaton, T. R. Eur. J. Org. Chem. 2011, 4965-4983.

(29) Tour, J. M.; Rawlett, A. M.; Kozaki, M.; Yao, Y.; Jagessar, R. C.; Dirk, S. M.; Price, D. W.; Reed, M. A.; Zhou, C.-W.; Chen, J.; Wang, W.; Campbell, I. *Chem.—Eur. J.* **2001**, *7*, 5118–5134.

(30) Huber, R.; Gonzalez, M. T.; Wu, S.; Langer, M.; Grunder, S.; Horhoiu, V.; Mayor, M.; Bryce, M. R.; Wang, C. S.; Jitchati, R.; Schonenberger, C.; Calame, M. J. Am. Chem. Soc. **2008**, 130, 1080– 1084.

(31) Kaliginedi, V.; Moreno-García, P.; Valkenier, H.; Hong, W.; García-Suárez, V. M.; Buiter, P.; Otten, J. L. H.; Hummelen, J. C.; Lambert, C. J.; Wandlowski, T. J. Am. Chem. Soc. **2012**, *134*, 5262–5275.

(32) Hu, W.; Jiang, J.; Nakashima, H.; Luo, Y.; Kashimura, Y.; Chen,
 K.-Q.; Shuai, Z.; Furukawa, K.; Lu, W.; Liu, Y.; Zhu, D.; Torimitsu, K.
 Phys. Rev. Lett. 2006, *96*, 027801.

(33) Lu, Q.; Liu, K.; Zhang, H.; Du, Z.; Wang, X.; Wang, F. ACS Nano **2009**, 3, 3861–3868.

(34) Lu, Q.; Yao, C.; Wang, X.; Wang, F. J. Phys. Chem. C 2012, 116, 17853-17861.

(35) Tour, J. M.; Jones, L., II.; Pearson, D. L.; Lamba, J. J. S.; Burgin, T. P.; Whitesides, G. W.; Allara, D. L.; Parikh, A. N.; Atre, S. V. J. Am. Chem. Soc. **1995**, *117*, 9529–9534.

(36) Valkenier, H. V. H.; Huisman, E. H.; van Hal, P. A.; de Leeuw, D. M.; Chiechi, R. C.; Hummelen, J. C. *J. Am. Chem. Soc.* **2011**, *133*, 4930–4939.

(37) Xu, B. Q.; Tao, N. J. J. Science 2003, 301, 1221-1223.

(38) Wang, C. S.; Batsanov, A. S.; Bryce, M. R.; Martin, S.; Nichols, R. J.; Higgins, S. J.; Garcia-Suarez, V. M.; Lambert, C. J. *J. Am. Chem. Soc.* **2009**, *131*, 15647–15654.

(39) Kamenetska, M.; Quek, S. Y.; Whalley, A. C.; Steigerwald, M. L.; Choi, H. J.; Louie, S. G.; Nuckolls, C.; Hybertsen, M. S.; Neaton, J. B.; Venkataraman, L. J. Am. Chem. Soc. **2010**, *132*, 6817–6821.

(40) Grunder, S.; Huber, R.; Horhoiu, V.; Gonzalez, M. T.; Schoenenberger, C.; Calame, M.; Mayor, M. J. Org. Chem. 2007, 72, 8337–8344.

(41) Hong, W.; Manrique, D. Z.; Moreno-García, P.; Gulcur, M.; Mishchenko, A.; Lambert, C. J.; Bryce, M. R.; Wandlowski, T. J. Am. Chem. Soc. **2011**, 134, 2292–2304.

(42) Sonogashira, K. J. Organomet. Chem. 2002, 653, 46.

(43) Dabdoub, M. J.; Dabdoub, V. B.; Lenardão, E. J. *Tetrahedron Lett.* 2001, 42, 1807–1809.

(44) Hong, W.; Valkenier, H.; Meszaros, G.; Manrique, D. Z.; Mishchenko, A.; Putz, A.; Garcia, P. M.; Lambert, C. J.; Hummelen, J. C.; Wandlowski, T. *Beilstein J. Nanotechnol.* **2011**, *2*, 699–713.

(45) Quek, S. Y.; Kamenetska, M.; Steigerwald, M. L.; Choi, H. J.; Louie, S. G.; Hybertsen, M. S.; Neaton, J. B.; Venkataraman, L. *Nat. Nanotechnol.* **2009**, *4*, 230–234.

(46) Newton, M. D; Smalley, J. F. Phys. Chem. Chem. Phys. 2007, 9, 555-572.

(47) Mishchenko, A.; Zotti, L. A.; Vonlanthen, D.; Burkle, M.; Pauly, F.; Cuevas, J. C.; Mayor, M.; Wandlowski, Th. J. Am. Chem. Soc. 2011, 133, 184–187.

(48) Martin, C. A.; Ding, D.; Sorensen, J. K.; Bjornholm, T.; van Ruitenbeek, J. M.; van der Zant, H. S. J. *J. Am. Chem. Soc.* **2008**, *130*, 13198–13199.

(49) Wold, D. J.; Haag, R.; Rampi, M. A.; Frisbie, C. D. J. Phys. Chem. B 2002, 106, 2813–2816.

(50) Ishida, T.; Mizutani, W.; Aya, Y.; Ogiso, H.; Sasaki, S.; Tokumoto, H. J. Phys. Chem. B **2002**, *106*, 5886–5892.

(51) Moreno-García, P.; Gulcur, M.; Manrique, D. Z.; Pope, Th.; Hong, W.; Kaliginedi, V.; Huang, C.; Batsanov, A. S.; Bryce, M. R.; Lambert, C.; Wandlowski, Th. J. Am. Chem. Soc. **2013**, 135, 12228– 12240.

(52) Liu, H.; Wang, N.; Zhao, J.; Guo, Y.; Yin, X.; Boey, F. Y. C.; Zhang, H. *ChemPhysChem* **2008**, *9*, 1416–1424.

(53) Yamada, R.; Kumazawa, H.; Noutoshi, T.; Tanaka, S.; Tada, H. Nano Lett. **2008**, 8, 1237–1240.

(54) Sedghi, G.; Garcia-Suarez, V. M.; Esdaile, L. J.; Anderson, H. L.; Lambert, C. J.; Martin, S.; Bethell, D.; Higgins, S. J.; Elliott, M.; Bennett, N.; Macdonald, E. J.; Nichols, R. J. *Nat. Nanotechnol.* **2011**, *6*, 517–523.

 (55) Yamada, R.; Kumazawa, H.; Tanaka, S.; Tada, H. Appl. Phys. Express 2009, 2, 025002.